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Continuation/Divisional Docket No.: 0941.64338
Parent Docket No.: 0941.63365

PATENT APPLICATION

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of: **Abarra et al.**

Application No.: 09/425,788

Filed: **October 22, 1999**

For: **MAGNETIC RECORDING MEDIUM**

AND MAGNETIC STORAGE APPARATUS

Art Unit: 1775

Examiner:

JC932 U.S. PRO
09/58923
06/06/00

ASSISTANT COMMISSIONER FOR PATENTS
WASHINGTON, D.C. 20231

NOTIFICATION OF FILING OF CONTINUATION-IN-PART APPLICATION

Notification is hereby being made of the filing of a:

☐ continuation

☒ continuation-in-part

☐ divisional

application for this case

☒ concurrently herewith.

☐ on _____
Date

Reg No. 29,367



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CERTIFICATION UNDER 37 C.F.R. 1.8(a) and 1.10

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Docket No.: 0941.64338

PATENT APPLICATION

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

Anticipated Classification of this application:

Class: Subclass:

Prior application: 09/425,788

Examiner:

Art Unit: 1775

TRANSMITTAL LETTER FOR FILING CONTINUATION-IN-PART APPLICATION

BOX PATENT APPLICATION
ASSISTANT COMMISSIONER FOR PATENTS
WASHINGTON, D.C. 20231

Sir:

This is a

☒ Continuation-in-part application,

☐ Divisional application,

under 37 CFR 1.53(b), of pending prior application Serial No. 09/425,788 filed on **October 22, 1999** for **MAGNETIC RECORDING MEDIUM AND MAGNETIC STORAGE APPARATUS.**

CERTIFICATION UNDER 37 CFR 1.10

I certify that, on the date shown below, these documents are being deposited with the United States Postal Service in an envelope addressed to the Assistant Commissioner of Patents, Washington, D.C. 20231, as "Express Mail Post Office to Addressee," Mailing Label No. **EL409492351US.**

Date June 6, 2000

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1. () Enclosed is a copy of the prior application, including the oath or declaration as originally filed. The attached application papers are a true copy of prior application Serial No.: ^^^^ as originally filed on ^^^^ , and no amendments referred to in the oath or declaration as originally filed introduced new matter. This statement is made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, under Section 1001 of Title 18 of the United States Code and that such willful false statements may jeopardize the validity of the application or any patent issuing thereon.

The copy of the papers of the prior application as filed are as follows:

- () ___ pages of specification including claims and abstract
 - () ___ sheets(s) of drawing
 - () ___ pages of declaration and power of attorney
2. () Enclosed is a replacement specification reflecting previously submitted amendments and/or additional formal amendments. No new matter is added by the changes. A copy of the oath or declaration as originally filed is also included.

The new application includes:

- (X) 27 pages of specification including claims and abstract
 - (X) 10 sheets(s) of drawing
 - (X) 4 pages of declaration and power of attorney
3. () A statement (Declaration) claiming small entity status was filed in the prior application and small entity status for this application is proper and desired.
4. () Cancel in this application original claims _____ of the prior application before calculating the filing fee. (At least one original independent claim must be retained for filing purposes).

5. ☒ The filing fee is calculated below:

CLAIMS AS FILED IN THE PRIOR APPLICATION, LESS ANY CLAIMS CANCELED BY AMENDMENT BELOW

For	No. Filed	No. Extra	Rate	Fee
Basic Fee...				\$690.00
Total Claims...	11 - 20 =	0	x \$18.00 =	\$ 0
Independent Claims...	4 - 3 =	1	x \$78.00 =	\$78.00
		Total Filing Fee		\$768.00
Statement of Status as Small Entity Reducing Filing Fee By Half To				\$ -0-

6. ☐ A check in the amount \$_____ is enclosed.
7. ☐ The Commissioner is hereby authorized to charge any additional fees which may be required for this application under 37 C.F.R. 1.16-1.17, or credit any overpayment, to Deposit Account No. 07-2069. A duplicate copy of this sheet is enclosed.
8. ☐ A petition and fee for an extension of time has been filed in the pending prior application. A copy of the petition is attached.
9. ☐ Amend the specification by inserting before the first line the sentence: -- This is a ☐ continuation, ☐ divisional, ☐ continuation-in-part, of application Serial No. _____, filed _____.--
10. ☒ Priority of application Serial No. **11-161329** filed on **June 8, 1999** in **Japan**, and Serial No. **2000-107075**, filed on **April 7, 2000**, in **Japan**, is claimed under 35 U.S.C. 119.
- ☒ The certified copy of application Serial No. **11-161329** was filed in prior application Serial No. **09/425,788**, filed **October 22, 1999**. The certified copy of Application Serial No. **2000-107075** will be filed at a later date.

11. (X) The prior application is assigned of record to: **Fujitsu Limited, Kawasaki-shi, Japan**. The assignment was recorded on **October 22, 1999** on Reel **10338**, Frame **0169**.
12. (X) The inventors of the invention of the parent application, serial number **09/425,788** are **E. Noel Abarra, Iwao Okamoto** and **Yoshifumi Mizoshita**.
13. (X) An information disclosure statement is submitted herewith.
14. () A preliminary amendment is enclosed herewith.
15. () Please delete the name(s) of _____ who are (is) not a (an) inventor(s) of the subject matter of the () continuing () divisional application.
16. (X) The power of attorney in the prior application is to:


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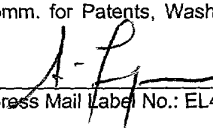
June 6, 2000

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SPECIFICATION

TO ALL WHOM IT MAY CONCERN:

BE IT KNOWN THAT WE, Iwao Okamoto, a citizen of Japan residing at Kawasaki-shi, Kanagawa, Japan, E. Noel Abarra, a citizen of Philippine residing at Kawasaki-shi, Kanagawa, Japan and Yoshifumi Mizoshita, a citizen of Japan residing at Kawasaki-shi, Kanagawa, Japan have invented certain new and useful improvements in

MAGNETIC RECORDING MEDIUM AND MAGNETIC STORAGE APPARATUS

of which the following is a specification : -

TITLE OF THE INVENTION

MAGNETIC RECORDING MEDIUM AND MAGNETIC
STORAGE APPARATUS

5 BACKGROUND OF THE INVENTION

1. Field of the Invention

This application is a Continuation-In-Part
Application of a United States Patent Application
Serial No.09/425,788 filed October 22, 1999.

10 The present invention generally relates to
magnetic recording media and magnetic storage
apparatuses, and more particularly to a magnetic
recording medium and a magnetic storage apparatus
which are suited for high-density recording.

15 2. Description of the Related Art

The recording density of longitudinal
magnetic recording media, such as magnetic disks,
has been increased considerably, due to the
reduction of medium noise and the development of
20 magnetoresistive and high-sensitivity spin-valve
heads. A typical magnetic recording medium is
comprised of a substrate, an underlayer, a magnetic
layer, and a protection layer which are successively
stacked in this order. The underlayer is made of Cr
25 or a Cr-based alloy, and the magnetic layer is made
of a Co-based alloy.

Various methods have been proposed to
reduce the medium noise. For example, Okamoto et
al., "Rigid Disk Medium For 5 Gbit/in² Recording",
30 AB-3, Intermag '96 Digest proposes decreasing the
grain size and size distribution of the magnetic
layer by reducing the magnetic layer thickness by
the proper use of an underlayer made of CrMo, and a
U.S. Patent No.5,693,426 proposes the use of an
35 underlayer made of NiAl. Further, Hosoe et al.,
"Experimental Study of Thermal Decay in High-Density
Magnetic Recording Media", IEEE Trans. Magn. Vol.33,

1528 (1997), for example, proposes the use of an underlayer made of CrTiB. The underlayers described above also promote c-axis orientation of the magnetic layer in a plane which increases the remanence magnetization and the thermal stability of written bits. In addition, proposals have been made to reduce the thickness of the magnetic layer, to increase the resolution or to decrease the width of transition between written bits. Furthermore, proposals have been made to decrease the exchange coupling between grains by promoting more Cr segregation in the magnetic layer which is made of the CoCr-based alloy.

However, as the grains of the magnetic layer become smaller and more magnetically isolated from each other, the written bits become unstable due to thermal activation and to demagnetizing fields which increase with linear density. Lu et al., "Thermal Instability at 10 Gbit/in² Magnetic Recording", IEEE Trans. Magn. Vol.30, 4230 (1994) demonstrated, by micromagnetic simulation, that exchange-decoupled grains having a diameter of 10 nm and ratio $K_u V / k_B T \sim 60$ in 400 kfc i di-bits are susceptible to significant thermal decay, where K_u denotes the magnetic anisotropy constant, V denotes the average magnetic grain volume, k_B denotes the Boltzmann constant, and T denotes the temperature. The ratio $K_u V / k_B T$ is also referred to as a thermal stability factor.

It has been reported in Abarra et al., "Thermal Stability of Narrow Track Bits in a 5 Gbit/in² Medium", IEEE Trans. Magn. Vol.33, 2995 (1997) that the presence of intergranular exchange interaction stabilizes written bits, by MFM studies of annealed 200 kfc i bits on a 5 Gbit/in² CoCrPtTa/CrMo medium. However, more grain decoupling is essential for recording densities of

20 Gbit/in² or greater.

The obvious solution has been to increase the magnetic anisotropy of the magnetic layer. But unfortunately, the increased magnetic anisotropy
5 places a great demand on the head write field which degrades the "overwrite" performance which is the ability to write over previously written data.

In addition, the coercivity of thermally unstable magnetic recording medium increases rapidly
10 with decreasing switching time, as reported in He et al., "High Speed Switching in Magnetic Recording Media", J. Magn. Magn. Mater. Vol.155, 6 (1996), for magnetic tape media, and in J. H. Richter, "Dynamic Coercivity Effects in Thin Film Media", IEEE Trans.
15 Magn. Vol.34, 1540 (1997), for magnetic disk media. Consequently, the adverse effects are introduced in the data rate, that is, how fast data can be written on the magnetic layer and the amount of head field required to reverse the magnetic grains.

On the other hand, another proposed method
20 of improving the thermal stability increases the orientation ratio of the magnetic layer, by appropriately texturing the substrate under the magnetic layer. For example, Akimoto et al.,
25 "Relationship Between Magnetic Circumferential Orientation and Magnetic Thermal Stability", J. Magn. Magn. Mater. (1999), in press, report through micromagnetic simulation, that the effective ratio $K_u V / k_B T$ is enhanced by a slight increase in the
30 orientation ratio. This further results in a weaker time dependence for the coercivity which improves the overwrite performance of the magnetic recording medium, as reported in Abarra et al., "The Effect of Orientation Ratio on the Dynamic Coercivity of Media
35 for >15 Gbit/in² Recording", EB-02, Intermag '99, Korea.

Furthermore, kept magnetic recording

media have been proposed for thermal stability improvement. The keeper layer is made up of a magnetically soft layer parallel to the magnetic layer. This soft layer can be disposed above or
5 below the magnetic layer. Oftentimes, a Cr isolation layer is interposed between the soft layer and the magnetic layer. The soft layer reduces the demagnetizing fields in written bits on the magnetic layer. However, coupling the magnetic layer to a
10 continuously-exchanged coupled soft layer defeats the purpose of decoupling the grains of the magnetic layer. As a result, the medium noise increases.

Various methods have been proposed to improve the thermal stability and to reduce the
15 medium noise. However, there was a problem in that the proposed methods do not provide a considerable improvement of the thermal stability of written bits, thereby making it difficult to greatly reduce the medium noise. In addition, there was another
20 problem in that some of the proposed methods introduce adverse effects on the performance of the magnetic recording medium due to the measures taken to reduce the medium noise.

More particularly, in order to obtain a
25 thermally stable performance of the magnetic recording medium, it is conceivable to (i) increase the magnetic anisotropy constant K_u , (ii) decrease the temperature T or, (iii) increase the grain volume V of the magnetic layer. However, measure
30 (i) increases the coercivity, thereby making it more difficult to write information on the magnetic layer. In addition, measure (ii) is impractical since in magnetic disk drives, for example, the operating temperature may become greater than 60°C .
35 Furthermore, measure (iii) increases the medium noise as described above. As an alternative for measure (iii), it is conceivable to increase the

thickness of the magnetic layer, but this would lead to deterioration of the resolution.

SUMMARY OF THE INVENTION

5 Accordingly, it is a general object of the present invention to provide a novel and useful magnetic recording medium and magnetic storage apparatus, in which the problems described above are eliminated.

10 Another and more specific object of the present invention is to provide a magnetic recording medium and a magnetic storage apparatus, which can improve the thermal stability of written bits without increasing the medium noise, so as to enable
15 a reliable high-density recording without introducing adverse effects on the performance of the magnetic recording medium, that is, unnecessarily increasing the magnetic anisotropy, and can improve the recording resolution by
20 improving the in-plane crystal orientation of the magnetic layer.

 Still another object of the present invention is to provide a magnetic recording medium characterized by at least one exchange layer
25 structure; and a magnetic layer formed on the exchange layer structure, the exchange layer structure comprising a ferromagnetic layer, and a non-magnetic coupling layer provided on the ferromagnetic layer and under the magnetic layer,
30 the ferromagnetic layer and the magnetic layer having antiparallel magnetizations, and the non-magnetic coupling layer being made of a Ru-M3 alloy, where M3 is an added element or alloy, and a lattice mismatch between the non-magnetic coupling layer and
35 the magnetic layer and the ferromagnetic layer respectively disposed above and below the non-magnetic coupling layer is adjusted to approximately

6% or less by addition of M3. According to the magnetic recording medium of the present invention, it is possible to provide a magnetic recording medium which can improve the thermal stability of written bits, so as to enable reliable high-density recording without degrading the overwrite performance, and to improve the recording resolution by improving the in-plane crystal orientation of the magnetic layer.

10 A further object of the present invention is to provide a magnetic recording medium characterized by at least one exchange layer structure; and a magnetic layer formed on the exchange layer structure, the exchange layer structure comprising a ferromagnetic layer, and a non-magnetic coupling layer provided on the ferromagnetic layer and under the magnetic layer, the ferromagnetic layer and the magnetic layer having antiparallel magnetizations, the non-magnetic coupling layer being made of a Ru-M3 alloy, where M3 = Co, Cr, Fe, Ni, Mn or alloys thereof. According to the magnetic recording medium of the present invention, it is possible to provide a magnetic recording medium which can improve the thermal stability of written bits, so as to enable reliable high-density recording without degrading the overwrite performance, and to improve the recording resolution by improving the in-plane crystal orientation of the magnetic layer.

30 Another object of the present invention is to provide a magnetic storage apparatus comprising at least one magnetic recording medium of any of the types described above. According to the magnetic storage apparatus of the present invention, it is possible to provide a magnetic recording medium which can improve the thermal stability of written bits, so as to enable reliable high-density

recording without degrading the overwrite performance, and to improve the recording resolution by improving the in-plane crystal orientation of the magnetic layer.

5 Other objects and further features of the present invention will be apparent from the following detailed description when read in conjunction with the accompanying drawings.

10 BRIEF DESCRIPTION OF THE DRAWINGS

 FIG. 1 is a cross sectional view showing an important part of a first embodiment of the magnetic recording medium according to the present invention;

15 FIG. 2 is a cross sectional view showing an important part of a second embodiment of the magnetic recording medium according to the present invention;

 FIG. 3 is a diagram showing an in-plane magnetization curve of a single CoPt layer having a thickness of 10 nm on a Si substrate;

 FIG. 4 is a diagram showing an in-plane magnetization curve of two CoPt layers separated by a Ru layer having a thickness of 0.8 nm;

25 FIG. 5 is a diagram showing an in-plane magnetization curve of two CoPt layers separated by a Ru layer having a thickness of 1.4 nm;

 FIG. 6 is a diagram showing an in-plane magnetization curve two CoCrPt layers separated by a Ru having a thickness of 0.8 nm;

30 FIG. 7 is a cross sectional view showing an important part of an embodiment of the magnetic storage apparatus according to the present invention;

35 FIG. 8 is a plan view showing the important part of the embodiment of the magnetic storage apparatus;

FIG. 9 is a cross sectional view showing an important part of a third embodiment of the magnetic recording medium according to the present invention;

5 FIG. 10 is a diagram showing a magnetization curve which is obtained when pure Ru is used for a non-magnetic coupling layer of the magnetic recording medium; and

10 FIG. 11 is a diagram showing a magnetization curve which is measured by a vertical Kerr loopier while applying a magnetic field in a perpendicular direction with respect to a sample surface.

15 DESCRIPTION OF THE PREFERRED EMBODIMENTS

A description will hereinafter be given of embodiments of the present invention, by referring to the drawings.

20 First, a description will be given of the operating principle of the present invention.

The present invention submits the use of layers with antiparallel magnetization structures. For example, S. S. P. Parkin, "Systematic Variation of the Strength and Oscillation Period of Indirect
25 Magnetic Exchange Coupling through the 3d, 4d, and 5d Transition Metals", Phys. Rev. Lett. Vol.67, 3598 (1991) describes several magnetic transition metals such as Co, Fe and Ni that are coupled through thin non-magnetic interlayers such as Ru and Rh. On the
30 other hand, a U.S. Patent No.5,701,223 proposes a spin-valve which employs the above described layers as laminated pinning layers to stabilize the sensor.

For a particular Ru or Ir layer thickness between two ferromagnetic layers, the magnetizations
35 can be made parallel or antiparallel. For example, for a structure made up of two ferromagnetic layers of different thickness with antiparallel

magnetizations, the effective grain size of a magnetic recording medium can be increased without significantly affecting the resolution. A signal amplitude reproduced from such a magnetic recording medium is reduced due to the opposite magnetizations, but this can be rectified by adding another layer of appropriate thickness and magnetization direction, under the laminated magnetic layer structure, to thereby cancel the effect of one of the layers. As a result, it is possible to increase the signal amplitude reproduced from the magnetic recording medium, and to also increase the effective grain volume. Thermally stable written bits can therefore be realized.

The present invention increases the thermal stability of written bits by exchange coupling the magnetic layer to another ferromagnetic layer with an opposite magnetization or, by a laminated ferrimagnetic structure. The ferromagnetic layer or the laminated ferrimagnetic structure is made up of exchange-decoupled grains as the magnetic layer. In other words, the present invention uses an exchange pinning ferromagnetic layer or a ferrimagnetic multilayer to improve the thermal stability performance of the magnetic recording medium.

FIG. 1 is a cross sectional view showing an important part of a first embodiment of a magnetic recording medium according to the present invention.

The magnetic recording medium includes a non-magnetic substrate 1, a first seed layer 2, a NiP layer 3, a second seed layer 4, an underlayer 5, a non-magnetic intermediate layer 6, a ferromagnetic layer 7, a non-magnetic coupling layer 8, a magnetic layer 9, a protection layer 10, and a lubricant layer 11 which are stacked in the order shown in FIG.

1.

For example, the non-magnetic substrate 1 is made of Al, Al alloy or glass. This non-magnetic substrate 1 may or may not be mechanically textured.

5 The first seed layer 2 is made of Cr or Ti, for example, especially in the case where the non-magnetic substrate 1 is made of glass. The NiP layer 3 is preferably oxidized and may or may not be mechanically textured. The second seed layer 4 is
10 provided to promote a (001) or a (112) texture of the underlayer 5 when using a B2 structure alloy such as NiAl and FeAl for the underlayer 5. The second seed layer 4 is made of an appropriate material similar to that of the first seed layer 2.

15 In a case where the magnetic recording medium is a magnetic disk, the mechanical texturing provided on the non-magnetic substrate 1 or the NiP layer 3 is made in a circumferential direction of the disk, that is, in a direction in which tracks of
20 the disk extend.

The non-magnetic intermediate layer 6 is provided to further promote epitaxy, narrow the grain distribution of the magnetic layer 9, and orient the anisotropy axes of the magnetic layer 9
25 along a plane parallel to the recording surface of the magnetic recording medium. This non-magnetic intermediate layer 6 is made of a hcp structure alloy such as CoCr-M, where M = B, Mo, Nb, Ta, W or alloys thereof, and has a thickness in a range of 1
30 to 5 nm.

The ferromagnetic layer 7 is made of Co, Ni, Fe, Co-based alloy, Ni-based alloy, Fe-based alloy or the like. In other words, alloys such as CoCrTa, CoCrPt and CoCrPt-M, where M = B, Mo, Nb, Ta,
35 W, Cu or alloys thereof may be used for the ferromagnetic layer 7. This ferromagnetic layer 7 has a thickness in a range of 2 to 10 nm. The non-

coupling magnetic layer 8 is made of Ru, Ir, Rh, Ru-based alloy, Ir-based alloy, Rh-based alloy or the like. This non-magnetic coupling layer 8 preferably has a thickness in a range of 0.4 to 1.0 nm, and preferably on the order of approximately 0.6 to 0.8 nm. For this particular thickness range of the non-magnetic coupling layer 8, the magnetizations of the ferromagnetic layer 7 and the magnetic layer 9 are antiparallel. The ferromagnetic layer 7 and the non-magnetic coupling layer 8 form an exchange layer structure.

For a ferromagnetic layer 7 made of a Fe-based alloy, Cr forms a better non-magnetic coupling layer 8. In this case, the Cr non-magnetic coupling layer 8 has an optimum thickness of approximately 1.8 nm.

The magnetic layer 9 is made of Co or a Co-based alloys such as CoCrTa, CoCrPt and CoCrPt-M, where M = B, Mo, Nb, Ta, W, Cu or alloys thereof. The magnetic layer 9 has a thickness in a range of 5 to 30 nm. Of course, the magnetic layer 9 is not limited to a single-layer structure, and a multi-layer structure may be used for the magnetic layer 9.

The protection layer 10 is made of C, for example. In addition, the lubricant layer 11 is made of an organic lubricant, for example, for use with a magnetic transducer such as a spin-valve head. The protection layer 10 and the lubricant layer 11 form a protection layer structure on the recording surface of the magnetic recording medium.

Obviously, the layer structure under the exchange layer structure is not limited to that shown in FIG. 1. For example, the underlayer 5 may be made of Cr or Cr-based alloy and formed to a thickness in a range of 5 to 40 nm on the substrate 1, and the exchange layer structure may be provided on this underlayer 5.

Next, a description will be given of a second embodiment of the magnetic recording medium according to the present invention.

FIG. 2 is a cross sectional view showing an important part of the second embodiment of the magnetic recording medium. In FIG. 2, those parts which are the same as those corresponding parts in FIG. 1 are designated by the same reference numerals, and a description thereof will be omitted.

10 In this second embodiment of the magnetic recording medium, the exchange layer structure includes two non-magnetic coupling layers 8 and 8-1, and two ferromagnetic layers 7 and 7-1, which form a ferrimagnetic multilayer. This arrangement
15 increases the effective magnetization and signal, since the magnetizations of the two non-magnetic coupling layers 8 and 8-1 cancel each other instead of a portion of the magnetic layer 9. As a result, the grain volume and thermal stability of
20 magnetization of the magnetic layer 9 are effectively increased. More bilayer structures made up of the pair of ferromagnetic layer and non-magnetic coupling layer may be provided additionally to increase the effective grain volume, as long as
25 the easy axis of magnetization are appropriately oriented for the subsequently provided layers.

The ferromagnetic layer 7-1 is made of a material similar to that of ferromagnetic layer 7, and has a thickness range selected similarly to the
30 ferromagnetic layer 7. In addition, the non-magnetic coupling layer 8-1 is made of a material similar to that of the non-magnetic coupling layer 8, and has a thickness range selected similarly to the non-magnetic coupling layer 8. Within the
35 ferromagnetic layers 7-1 and 7, the c-axes are preferably in-plane and the grain growth columnar.

In this embodiment, the magnetic

anisotropy of the ferromagnetic layer 7-1 is preferably higher than that of the ferromagnetic layer 7. However, the magnetic anisotropy of the ferromagnetic layer 7-1 may be the same as or, be
5 higher than that of, the magnetic layer 9.

Furthermore, a remanent magnetization and thickness product of the ferromagnetic layer 7 may be smaller than that of the ferromagnetic layer 7-1.

FIG. 3 is a diagram showing an in-plane
10 magnetization curve of a single CoPt layer having a thickness of 10 nm on a Si substrate. In FIG. 3, the ordinate indicates the magnetization (emu), and the abscissa indicates the magnetic field (Oe). Conventional magnetic recording media show a
15 behavior similar to that shown in FIG. 3.

FIG. 4 is a diagram showing an in-plane magnetization curve of two CoPt layers separated by a Ru layer having a thickness of 0.8 nm, as in the case of the first embodiment of the magnetic
20 recording medium. In FIG. 4, the ordinate indicates the magnetization (Gauss), and the abscissa indicates the magnetic field (Oe). As may be seen from FIG. 4, the loop shows shifts near the magnetic field which indicate the antiparallel coupling.

FIG. 5 is a diagram showing an in-plane magnetization curve of two CoPt layers separated by a Ru layer having a thickness of 1.4 nm. In FIG. 5, the ordinate indicates the magnetization (emu), and the abscissa indicates the magnetic field (Oe). As
25 may be seen from FIG. 5, the magnetizations of the two CoPt layers are parallel.
30

FIG. 6 is a diagram showing an in-plane magnetization curve for two CoCrPt layers separated by a Ru having a thickness of 0.8 nm, as in the case
35 of the second embodiment of the magnetic recording medium. In FIG. 6, the ordinate indicates the magnetization (emu/cc), and the abscissa indicates

the field (Oe). As may be seen from FIG. 6, the loop shows shifts near the field which indicate the antiparallel coupling.

From FIGS. 3 and 4, it may be seen that
5 the antiparallel coupling can be obtained by the provision of the exchange layer structure. In addition, it may be seen by comparing FIG. 5 with FIGS. 4 and 6, the non-magnetic coupling layer 8 is desirably in the range of 0.4 to 0.9 nm in order to
10 achieve the antiparallel coupling.

Therefore, according to the first and second embodiments of the magnetic recording medium, it is possible to effectively increase the apparent grain volume of the magnetic layer by the exchange
15 coupling provided between the magnetic layer and the ferromagnetic layer via the non-magnetic coupling layer, without sacrificing the resolution. In other words, the apparent thickness of the magnetic layer is increased with regard to the grain volume of the
20 magnetic layer so that a thermally stable medium can be obtained, and in addition, the effective thickness of the magnetic layer is maintained since cancellation of signals especially from the bottom layers is achieved. This allows higher linear
25 density recording that is otherwise not possible for thick media. As a result, it is possible to obtain a magnetic recording medium with reduced medium noise and thermally stable performance.

Next, a description will be given of an
30 embodiment of a magnetic storage apparatus according to the present invention, by referring to FIGS. 7 and 8. FIG. 7 is a cross sectional view showing an important part of this embodiment of the magnetic storage apparatus, and FIG. 8 is a plan view showing
35 the important part of this embodiment of the magnetic storage apparatus.

As shown in FIGS. 7 and 8, the magnetic

storage apparatus generally includes a housing 13. A motor 14, a hub 15, a plurality of magnetic recording media 16, a plurality of recording and reproducing heads 17, a plurality of suspensions 18, 5 a plurality of arms 19, and an actuator unit 20 are provided within the housing 13. The magnetic recording media 16 are mounted on the hub 15 which is rotated by the motor 14. The recording and reproducing head 17 is made up of a reproducing head 10 such as a MR or GMR head, and a recording head such as an inductive head. Each recording and reproducing head 17 is mounted on the tip end of a corresponding arm 19 via the suspension 18. The arms 19 are moved by the actuator unit 20. The 15 basic construction of this magnetic storage apparatus is known, and a detailed description thereof will be omitted in this specification.

This embodiment of the magnetic storage apparatus is characterized by the magnetic recording 20 media 16. Each magnetic recording medium 16 has the structure of the first or second embodiment of the magnetic recording medium described above in conjunction with FIGS. 1 and 2. Of course, the number of magnetic recording media 16 is not limited 25 to three, and only one, two or four or more magnetic recording media 16 may be provided.

The basic construction of the magnetic storage unit is not limited to that shown in FIGS. 7 and 8. In addition, the magnetic recording medium 30 used in the present invention is not limited to a magnetic disk.

In the magnetic recording medium having the exchange layer structure such as that of the first embodiment shown in FIG. 1, when Ru is used 35 for the non-magnetic coupling layer 8 and a CoCr-based alloy is used for the magnetic layer 9, both of these layers 8 and 9 have the hcp structure. In

order to increase both the coercivity and resolution of the magnetic recording medium, it is desirable that the c-axis of the hcp structure is parallel with respect to the surface of the substrate 1. In
5 a case where a CoCr-based alloy is used for the ferromagnetic layer 7, the ferromagnetic layer 7 is grown epitaxially on the non-magnetic intermediate layer 6 which is made of an alloy having the hcp structure and oriented on the (002) face, and thus,
10 the in-plane orientation of the c-axis of the ferromagnetic layer 7 is very satisfactory.

On the other hand, the Ru which is used for the non-magnetic coupling layer 8 has the hcp structure, similarly as in the case of the CoCr-
15 based alloy, but the lattice constant of Ru is approximately 5% larger than the lattice constant of the CoCr-based alloy. For this reason, the epitaxial growth may be slightly obstructed due to the lattice mismatch between the ferromagnetic layer
20 7 and the non-magnetic coupling layer 8 or, between the non-magnetic coupling layer 8 and the magnetic layer 9. If the epitaxial growth is slightly obstructed due to the lattice mismatch, the coercivity of the magnetic recording medium
25 decreases, and the in-plane orientation of the c-axis of the CoCr-based alloy becomes unstable.

Next, a description will be given of an embodiment which can improve the epitaxial growth between the Ru and the CoCr-based alloy,
30 simultaneously increase the coercivity of the magnetic recording medium and improve the in-plane orientation of the c-axis of the CoCr-based alloy, and improve mainly the recording resolution characteristic of the magnetic recording medium.

35 FIG. 9 is a cross sectional view showing an important part of a third embodiment of the magnetic recording medium according to the present

invention.

The magnetic recording medium includes a non-magnetic substrate 16, a seed layer 17, an underlayer 18 made of a Cr-based alloy, a non-magnetic intermediate layer 19, a ferromagnetic layer 20, a non-magnetic coupling layer 21, a magnetic layer 22, a protection layer 23, and a lubricant layer 24 which are stacked in this order as shown in FIG. 9.

The non-magnetic substrate 16 is made of an Al alloy or glass, for example. The non-magnetic substrate 16 may or may not be mechanically textured. The seed layer 17 is made of NiP which is plated in a case where the non-magnetic substrate 16 is made of the Al alloy. The NiP seed layer 17 may or may not be mechanically textured. On the other hand, in a case where the non-magnetic substrate 16 is made of glass, the seed layer 17 is made of an intermetallic compound material having the B2 structure, such as NiAl and FeAl.

The non-magnetic intermediate layer 19 is provided to promote the epitaxial growth of the magnetic layer 22, reduction of the grain size distribution with of the magnetic layer 22, and the anisotropic axis (c-axis, axis of easy magnetization) orientation of the magnetic layer 22 along a plane parallel to the recording surface of the magnetic recording medium. The non-magnetic intermediate layer 19 is made of an alloy having the hcp structure, such as CoCr-M1, and has a thickness in a range of approximately 1 to 5 nm, where M1 = B, Mo, Bn, Ta, W or alloys thereof.

The ferromagnetic layer 20 is made of a material selected from a group of Co, Ni, Fe, Co-based alloys, Ni-based alloys, Fe-based alloys and the like. In other words, Co-based alloys including CoCrTa, CoCrPt and CoCrPt-M2 may be used for the

ferromagnetic layer 20, where M2 = B, Mo, Nb, Ta, W or alloys thereof. The ferromagnetic layer 20 has a thickness in a range of approximately 2 to 10 nm.

The non-magnetic coupling layer 21 is made
5 of an alloy having the hcp structure, such as Ru-M3, where M3 = Co, Cr, Fe, Ni, Mn or alloys thereof. For example, the non-magnetic coupling layer 21 has a thickness in a range of approximately 0.4 to 1.0 nm, and preferably in a range of approximately 0.6
10 to 0.8 nm. By setting the thickness of the non-magnetic coupling layer 21 within such a range, the magnetizations of the ferromagnetic layer 20 and the magnetic layer 22 become antiparallel. Accordingly, the ferromagnetic layer 20 and the non-magnetic
15 coupling layer 19 form an exchange layer structure.

The magnetic layer 22 is made of a material selected from a group of Co and Co-based alloys including CoCrTa, CoCrPt, and CoCrPt-M4, where M4 = B, Mo, Nb, Ta, W or alloys thereof. The
20 magnetic layer 22 has a thickness in a range of approximately 5 to 30 nm. Of course, the magnetic layer 22 is not limited to a single-layer structure, and a multi-layer structure may be used for the magnetic layer 22.

25 The protection layer 23 is made of C or diamond-like C (DLC). In addition, the lubricant layer 24 is made of an organic lubricant agent, in order to enable the magnetic recording medium to be used with a magnetic transducer such as a spin valve
30 head. The protection layer 23 and the lubricant layer 24 form a protection layer structure of the magnetic recording medium.

As described above, the non-magnetic coupling layer 21 is made of an alloy Ru-M3, where
35 M3 = Co, Cr, Fe, Ni, Mn or alloys thereof. In this embodiment, an amount of the element M3 added to the Ru is set within the following composition ranges so

as to maintain a stable hcp structure. In the following composition ranges, the numerical values following the brackets respectively indicate the amount in atomic percent (at%).

- 5
- Ru-Co(0 to 50 at%)
 - Ru-Cr(0 to 50 at%)
 - Ru-Fe(0 to 60 at%)
 - Ru-Ni(0 to 10 at%)
 - 10 Ru-Mn(0 to 50 at%)

FIG. 10 is a diagram showing a magnetization curve which is obtained when pure Ru is used for the non-magnetic coupling layer 21 of the magnetic recording medium shown in FIG. 9. In FIG. 10, the ordinate indicates the magnetization M (arbitrary units), and the abscissa indicates the magnetic field H (kOe). The magnetization curve shown in FIG. 10 was measured by a vibrating sample type magnetometer while applying a magnetic field parallel to the sample surface, that is, parallel to the recording surface of the magnetic recording medium. The magnetization curve has a constricted portion because of the existence of a region where the ferromagnetic layer 20 and the magnetic layer 22 form an antiparallel coupling.

In addition, a magnetization curve which is obtained when a Ru-M3 alloy is used for the non-magnetic coupling layer 21 was also measured similarly to the above. In the case where the Ru-M3 alloy is used for the non-magnetic coupling layer 21, it was also confirmed that a constricted portion is formed in the magnetization curve, similarly as in the case shown in FIG. 10, due to the existence of the region where the ferromagnetic layer 20 and the magnetic layer 22 form the antiparallel coupling.

In the first and fourth quadrants in FIG.

10, a linear portion of the magnetization curve on the higher magnetic field side of the constricted portion is extrapolated to the magnetic field axis, and the intersection with the magnetic field axis is defined as an in-plane coercivity $H_{c//}$.

FIG. 11 is a diagram showing a magnetization curve which is measured by a vertical Kerr loop (or loop) while applying a magnetic field in a perpendicular direction with respect to the sample surface, with respect to the magnetic recording medium for which the data shown in FIG. 10 were measured. In FIG. 11, the ordinate indicates the Kerr rotation (degrees), and the abscissa indicates the perpendicular magnetic field (Oe). A definition of a perpendicular coercivity $H_{c\perp}$ is shown in FIG. 11.

The extent of the in-plane orientation of the axis of easy magnetization of the magnetic layer 22 can be evaluated by a ratio $(H_{c\perp})/(H_{c//})$. The smaller this ratio $(H_{c\perp})/(H_{c//})$, the better the in-plane orientation of the magnetic layer 22.

Measured results of the in-plane coercivity $H_{c//}$ and the ratio $(H_{c\perp})/(H_{c//})$ for various materials used for the non-magnetic coupling layer 21 are shown in the following. In the following, the in-plane coercivity $H_{c//}$ of the various materials is indicated by a relative value relative to the in-plane coercivity $H_{c//} = 1$ for the case where pure Ru is used for the non-magnetic coupling layer 21.

Non-Magnetic Coupling Layer 21 (Relative Value)		Hc// (Hc \perp)/(Hc//)
	Ru	1 0.33
	Ru-Co(20at%)	1.10 0.23
5	Ru-Cr(20at%)	1.05 0.25
	Ru-Fe(20at%)	1.07 0.28
	Ru-Mn(20at%)	0.96 0.30
	Ru-Ni(10at%)	0.94 0.30

10 Therefore, it was confirmed that the
ratios (Hc \perp)/(Hc//) for the cases where the Ru-M3
alloys are used for the non-magnetic coupling layer
21 are improved according to this embodiment, as
15 compared to the case where pure Ru is used for the
non-magnetic coupling layer 21. As a result, it was
confirmed that the recording resolution is improved
by approximately 1.5 to 2.5 % by the improved in-
plane orientation of the magnetic layer 22.

A lattice mismatch of the intervals of the
20 (002) faces of the hcp structure of the Ru used for
the non-magnetic coupling layer 21 with respect to
the magnetic layer 22 and the ferromagnetic layer 20
respectively disposed above and below the non-
magnetic coupling layer 21 is normally approximately
25 5 % at the maximum and approximately 8 % in a worst
case. But by the addition of the element M3 to the
Ru, it was confirmed that the lattice mismatch can
be reduced to approximately 6 % or less, and
preferably approximately 2 % or less. Furthermore,
30 the element M3 added to the Ru is preferably Co, Cr,
Fe, Ni, Mn or alloys thereof, but it is of course
possible to adjust the lattice mismatch by adding to
the Ru a material selected from a group of Ir, Mo,
Nb, Pt, Rh, Ta, Ti, V, W and alloys thereof.

35 Of course, this embodiment may be applied
similarly to the construction of the second
embodiment of the magnetic recording medium

described above.

Further, the present invention is not
limited to these embodiments, but various variations
and modifications may be made without departing from
5 the scope of the present invention.

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WHAT IS CLAIMED IS

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1. A magnetic recording medium
comprising:

at least one exchange layer structure; and
a magnetic layer formed on said exchange layer
10 structure,

said exchange layer structure comprising a
ferromagnetic layer, and a non-magnetic coupling
layer provided on said ferromagnetic layer and under
said magnetic layer,

15 said ferromagnetic layer and said magnetic
layer having antiparallel magnetizations,

said non-magnetic coupling layer being made of
a Ru-M3 alloy, where M3 is an added element or alloy,
and a lattice mismatch between said non-magnetic
20 coupling layer and said magnetic layer and said
ferromagnetic layer respectively disposed above and
below said non-magnetic coupling layer is adjusted
to approximately 6% or less by addition of M3.

25

2. The magnetic recording medium as
claimed in claim 1, wherein said non-magnetic
30 coupling layer has a thickness in a range of 0.4 to
1.0 nm.

35

3. The magnetic recording medium as
claimed in claim 1, wherein said ferromagnetic layer

is made of a material selected from a group of Co, Ni, Fe, Ni-based alloys, Fe-based alloys, and Co-based alloys including CoCrTa, CoCrPt and CoCrPt-M2, where M2 = B, Mo, Nb, Ta, W, Cu or alloys thereof.

5

4. The magnetic recording medium as
10 claimed in claim 1, wherein said magnetic layer is made of a material selected from a group of Co, and Co-based alloys including CoCrTa, CoCrPt and CoCrPt-M4, where M4 = B, Mo, Nb, Ta, W, Cu or alloys thereof.

15

5. A magnetic recording medium
20 comprising:
at least one exchange layer structure; and
a magnetic layer formed on said exchange layer structure,
said exchange layer structure comprising a
25 ferromagnetic layer, and a non-magnetic coupling layer provided on said ferromagnetic layer and under said magnetic layer,
said ferromagnetic layer and said magnetic layer having antiparallel magnetizations,
30 said non-magnetic coupling layer being made of a Ru-M3 alloy, where M3 = Co, Cr, Fe, Ni, Mn or alloys thereof.

35

6. The magnetic recording medium as

claimed in claim 5, wherein an amount of the element M3 added to Ru is 50 at% or less for Co, 50 at% or less for Cr, 60 at% or less for Fe, 10 at% or less for Ni, and 50 at% or less for Mn.

5

7. The magnetic recording medium as
10 claimed in claim 5, wherein said non-magnetic coupling layer has a thickness in a range of 0.4 to 1.0 nm.

15

8. The magnetic recording medium as
claimed in claim 5, wherein said ferromagnetic layer
is made of a material selected from a group of Co,
20 Ni, Fe, Ni-based alloys, Fe-based alloys, and Co-based alloys including CoCrTa, CoCrPt and CoCrPt-M2, where M2 = B, Mo, Nb, Ta, W, Cu or alloys thereof.

25

9. The magnetic recording medium as
claimed in claim 5, wherein said magnetic layer is
made of a material selected from a group of Co, and
30 Co-based alloys including CoCrTa, CoCrPt and CoCrPt-M4, where M4 = B, Mo, Nb, Ta, W, Cu or alloys thereof.

35

10. A magnetic storage apparatus

comprising:

at least one magnetic recording medium
comprising at least one exchange layer structure,
and a magnetic layer formed on said exchange layer
5 structure,

said exchange layer structure comprising a
ferromagnetic layer, and a non-magnetic coupling
layer provided on said ferromagnetic layer and under
said magnetic layer,

10 said ferromagnetic layer and said magnetic
layer having antiparallel magnetizations,

said non-magnetic coupling layer being made of
a Ru-M3 alloy, where M3 is an added element or alloy,
and a lattice mismatch between said non-magnetic
15 coupling layer and said magnetic layer and said
ferromagnetic layer respectively disposed above and
below said non-magnetic coupling layer is adjusted
to approximately 6% or less by addition of M3.

20

11. A magnetic storage apparatus
comprising:

25 at least one magnetic recording medium
comprising at least one exchange layer structure,
and a magnetic layer formed on said exchange layer
structure,

said exchange layer structure comprising a
30 ferromagnetic layer, and a non-magnetic coupling
layer provided on said ferromagnetic layer and under
said magnetic layer,

said ferromagnetic layer and said magnetic
layer having antiparallel magnetizations,

35 said non-magnetic coupling layer being made of
a Ru-M3 alloy, where M3 = Co, Cr, Fe, Ni, Mn or
alloys thereof.

ABSTRACT OF THE DISCLOSURE

A magnetic recording medium is constructed to include at least one exchange layer structure, and a magnetic layer formed on the exchange layer structure, where the exchange layer structure comprises a ferromagnetic layer, and a non-magnetic coupling layer provided on the ferromagnetic layer and under the magnetic layer. The ferromagnetic layer and the magnetic layer have antiparallel magnetizations. The non-magnetic coupling layer is made of a Ru-M3 alloy, where M3 is an added element or alloy, and a lattice mismatch between the non-magnetic coupling layer and the magnetic layer and the ferromagnetic layer respectively disposed above and below the non-magnetic coupling layer is adjusted to approximately 6% or less by addition of M3.

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FIG.1

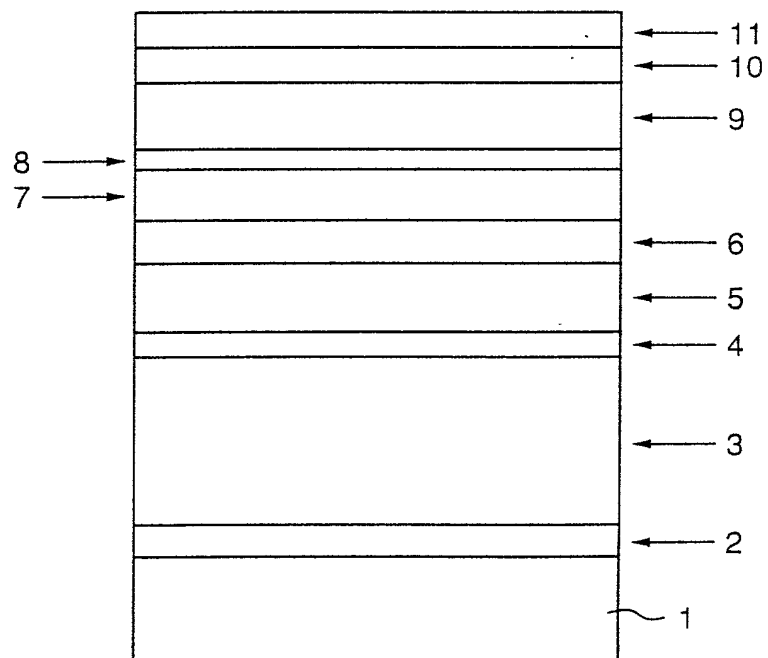


FIG. 2

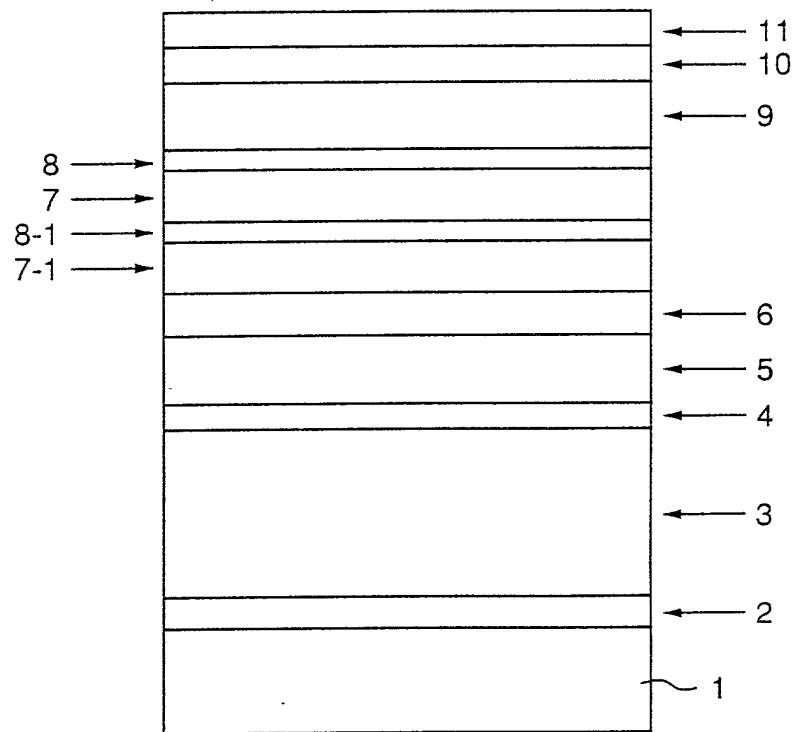


FIG.3

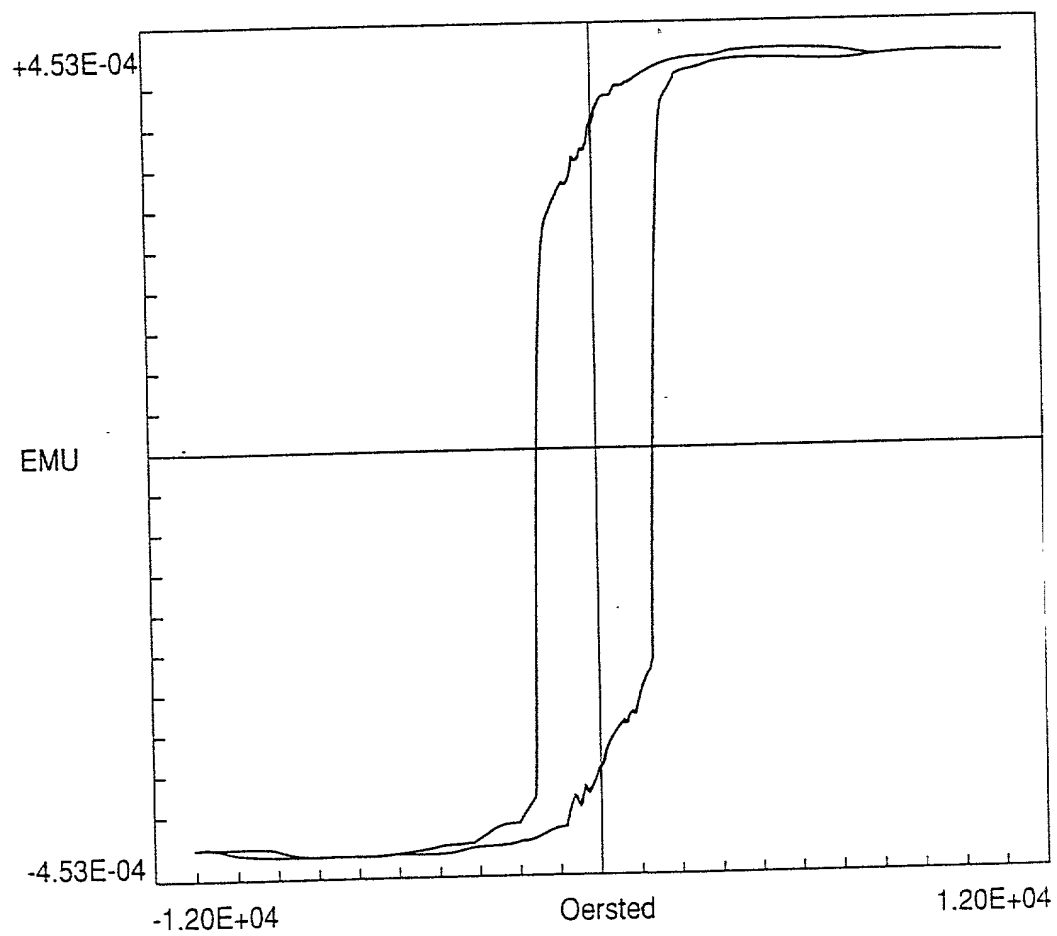


FIG.4

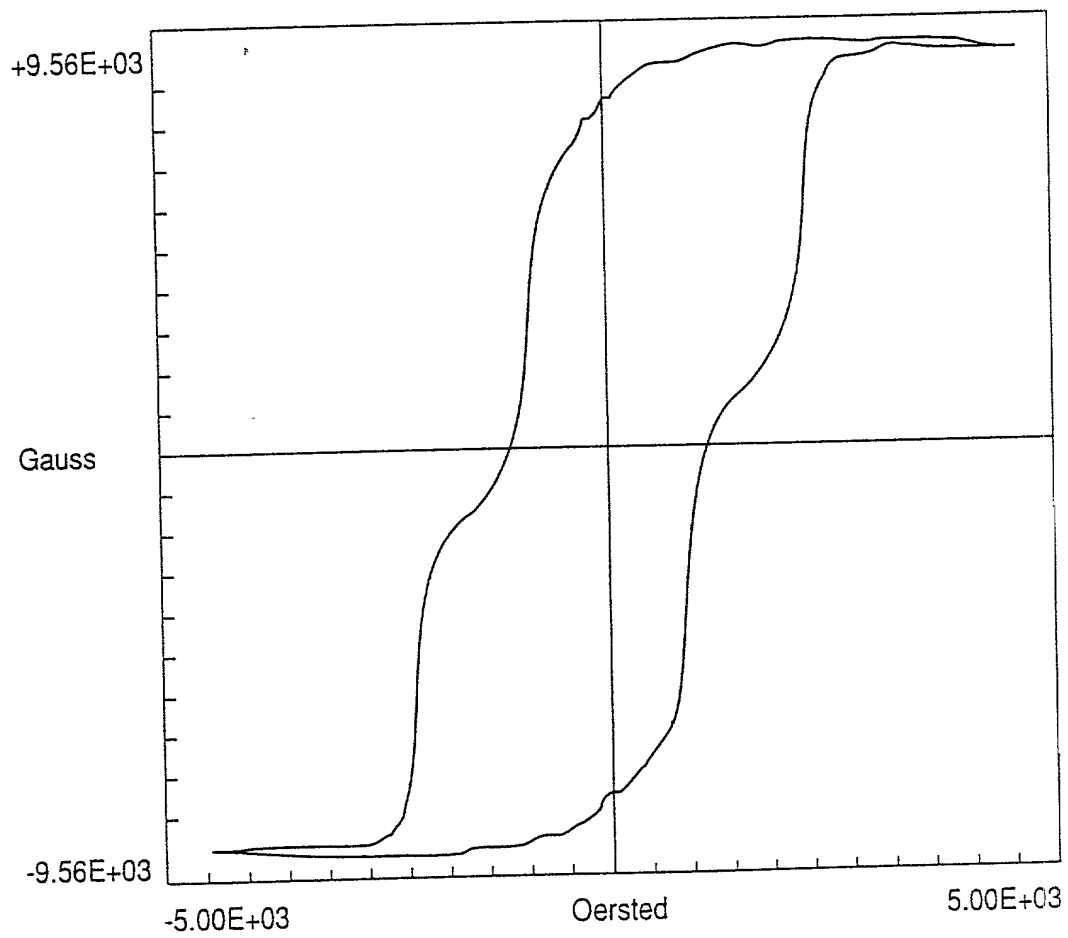


FIG.5

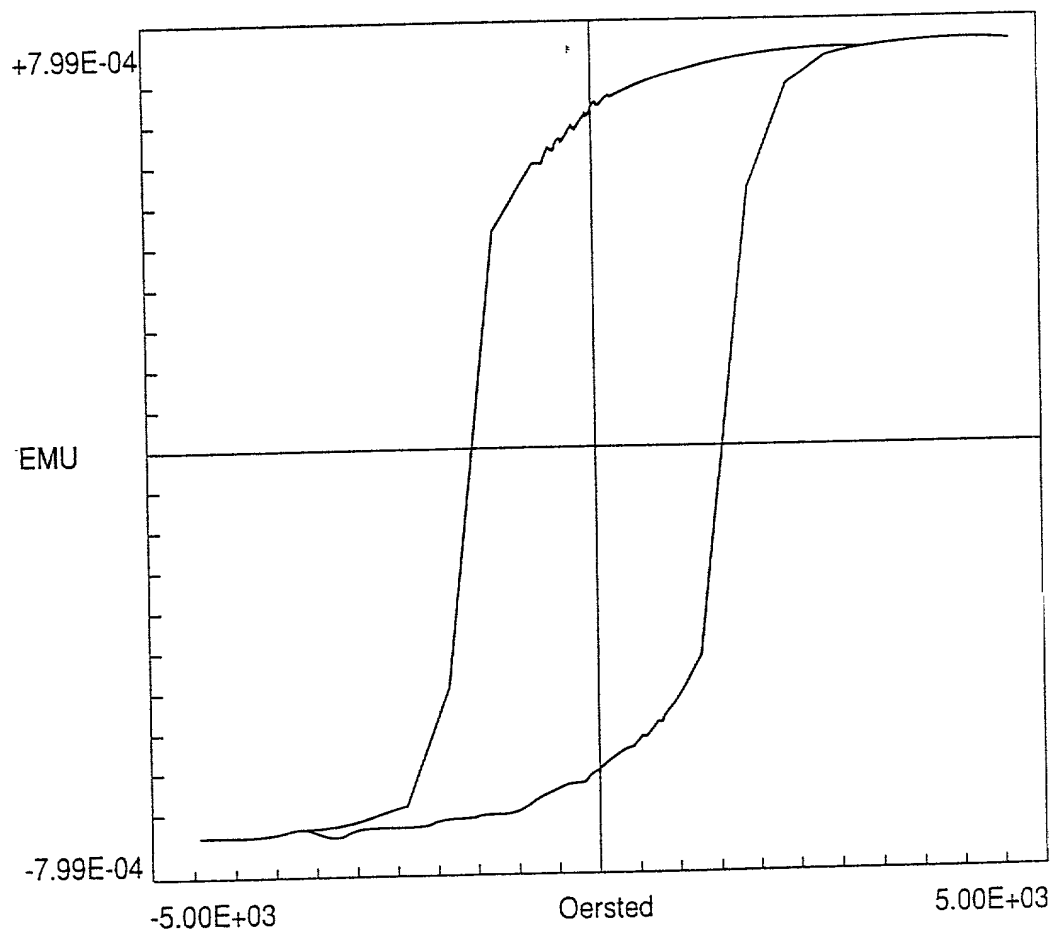


FIG.6

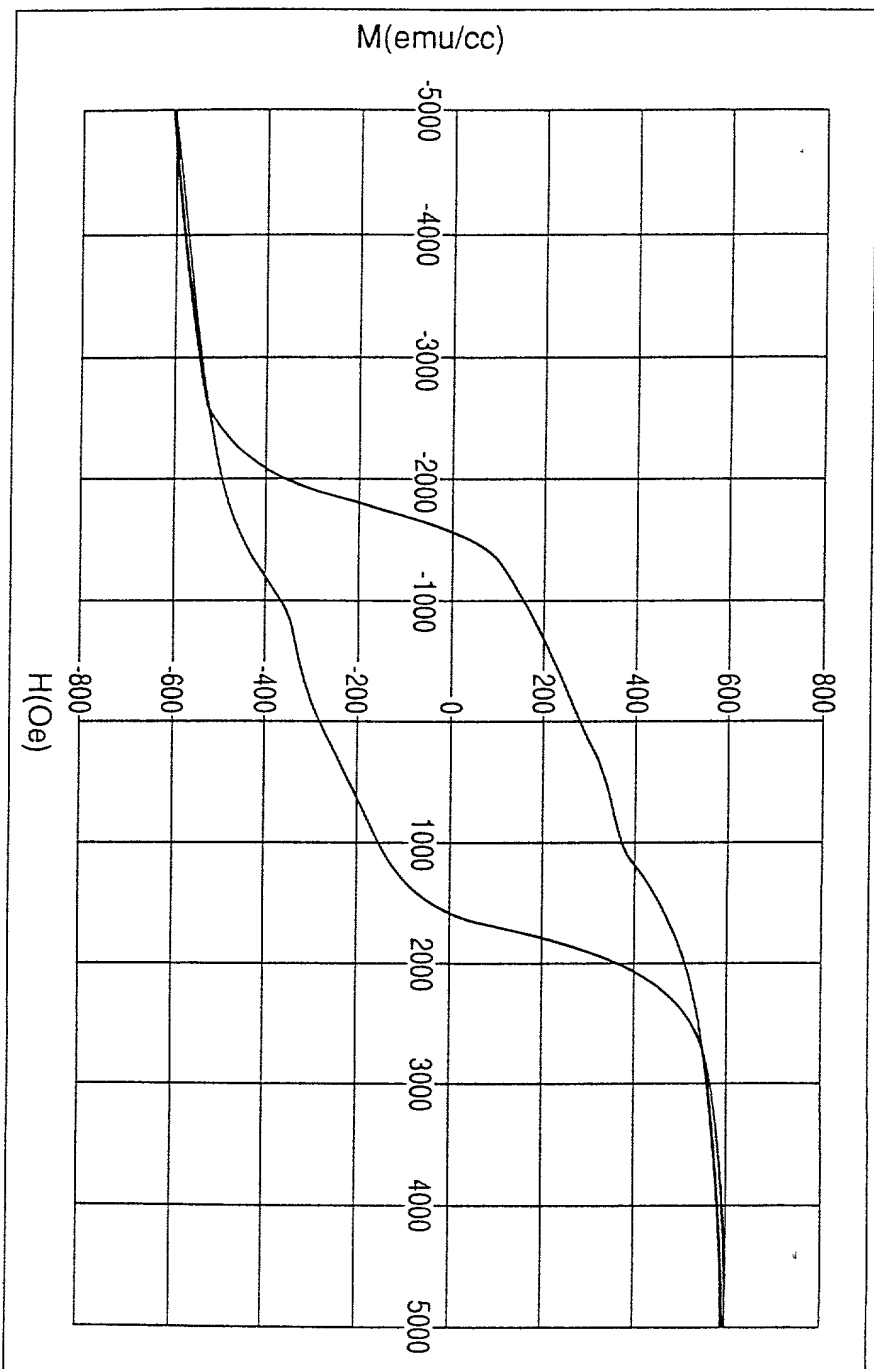


FIG.7

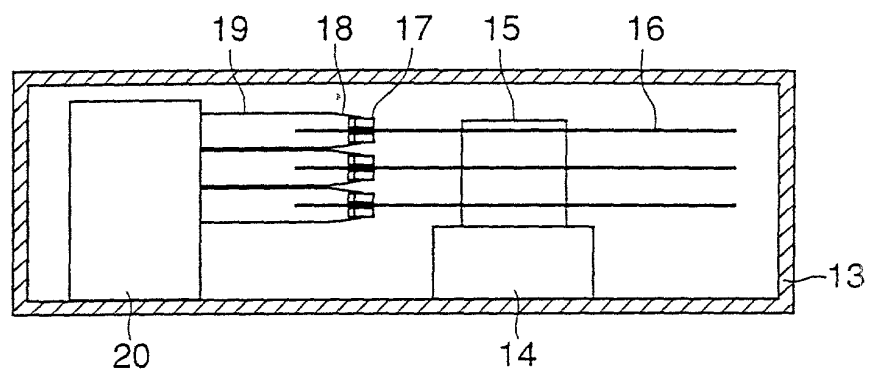


FIG.8

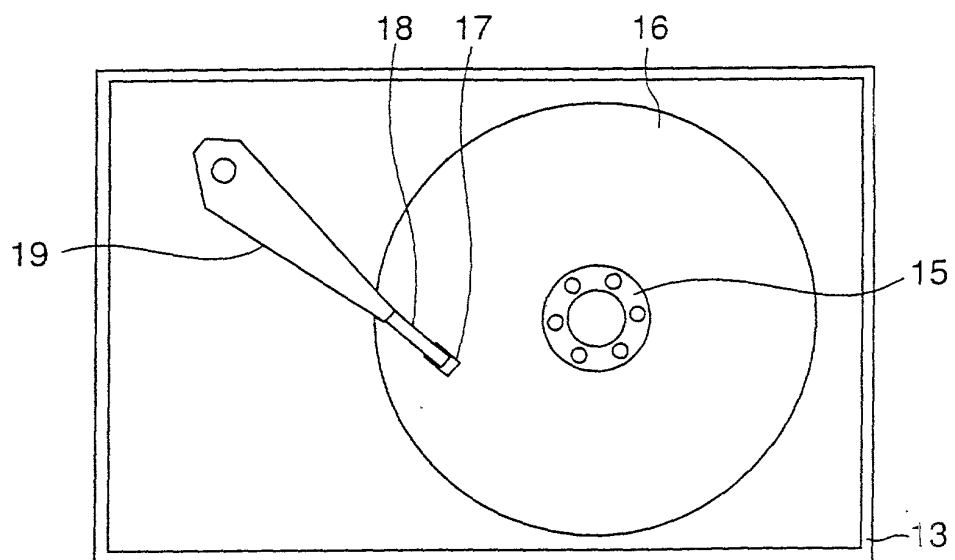


FIG.9

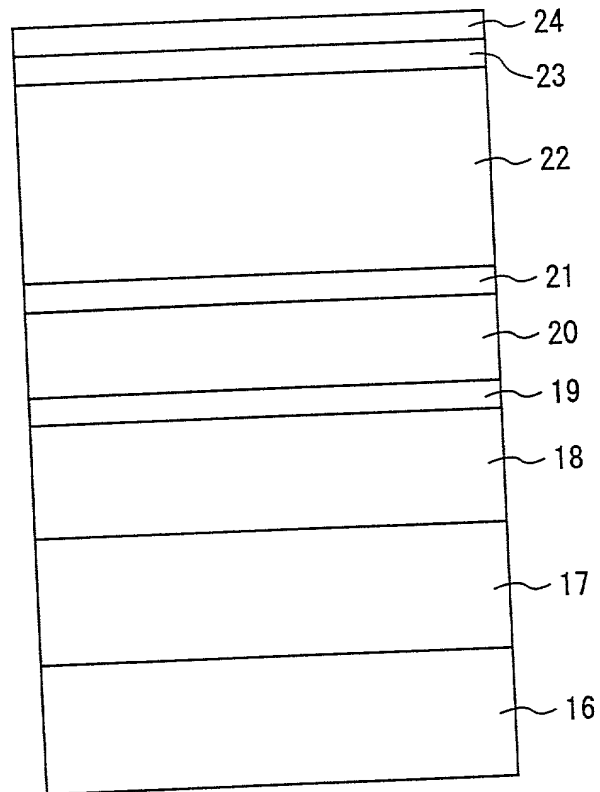


FIG.10

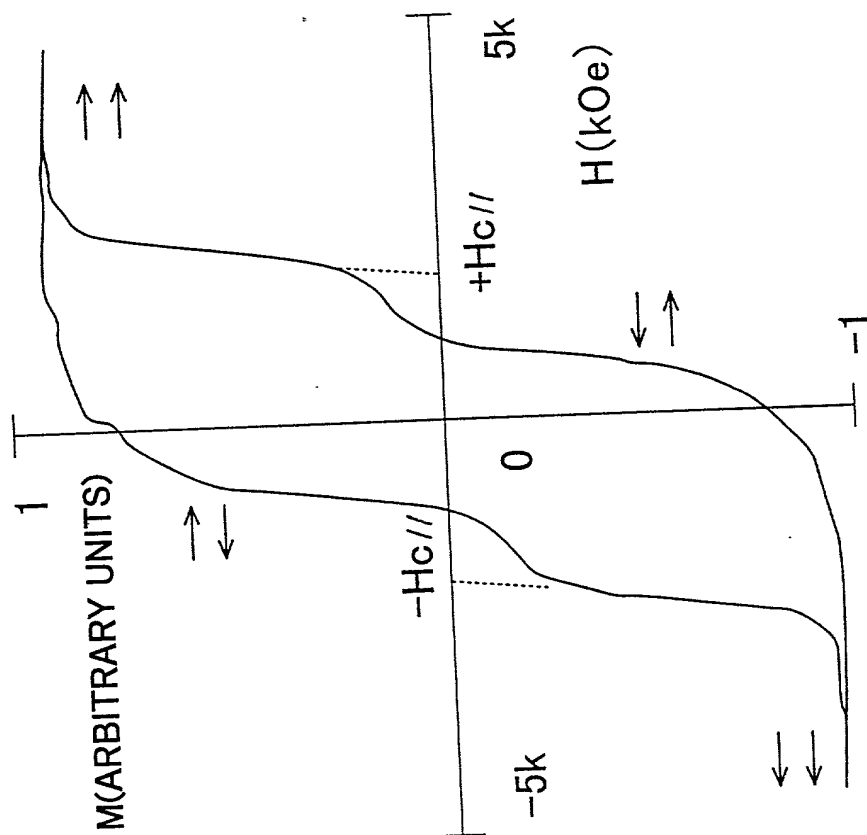
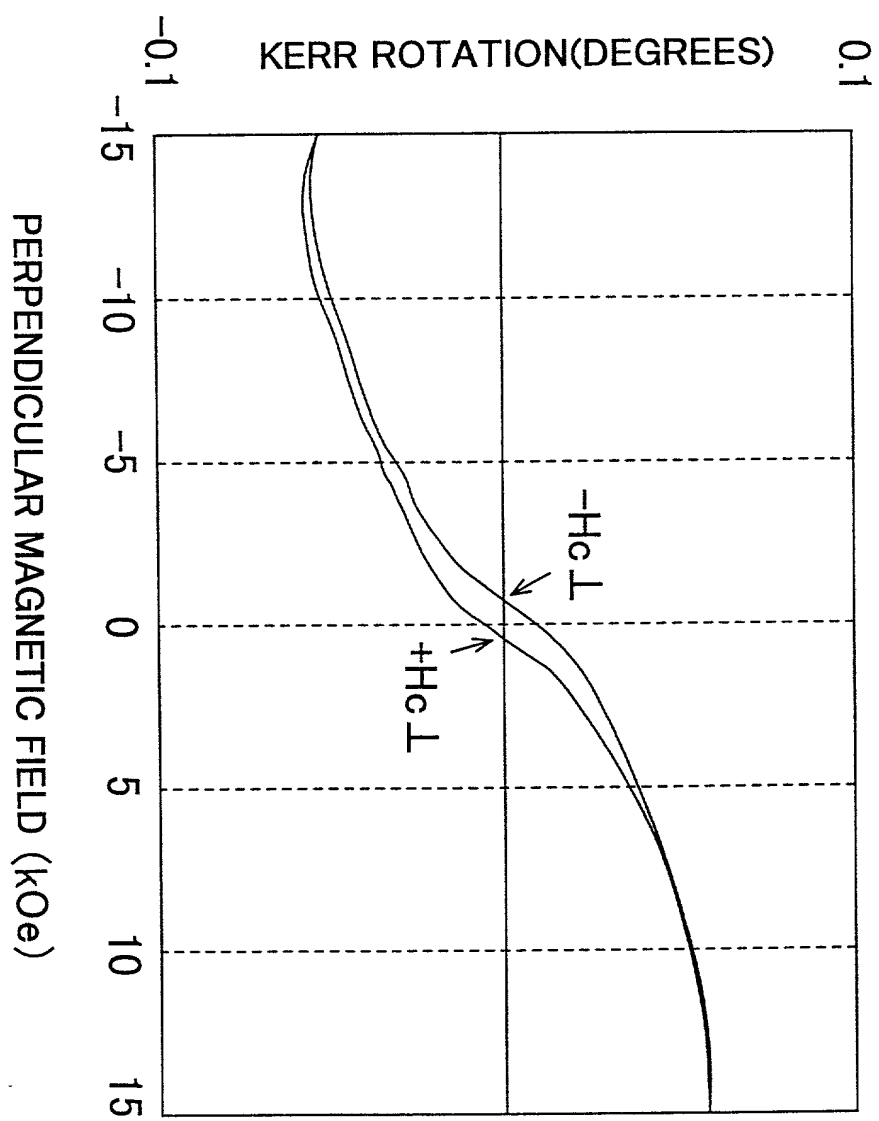


FIG.11



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Declaration and Power of Attorney For Patent Application

特許出願宣言書及び委任状

Japanese Language Declaration

日本語宣言書

下記の氏名の発明者として、私は以下の通り宣言します。

As a below named inventor, I hereby declare that:

私の住所、私書箱、国籍は下記の私の氏名の後に記載された通りです。

My residence, post office address and citizenship are as stated next to my name.

下記の名称の発明に関して請求範囲に記載され、特許出願している発明内容について、私が最初かつ唯一の発明者（下記の氏名が一つの場合）もしくは最初かつ共同発明者であると（下記の名称が複数の場合）信じています。

I believe I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled

MAGNETIC RECORDING MEDIUM AND

MAGNETIC STORAGE APPARATUS

上記発明の明細書（下記の欄でx印がついていない場合は、本書に添付）は、

the specification of which is attached hereto unless the following box is checked:

☐ __月__日に提出され、米国出願番号または特許協定条約国際出願番号を____とし、
（該当する場合）____に訂正されました。

☐ was filed on _____
as United States Application Number or
PCT International Application Number
_____ and was amended on
_____ (if applicable).

私は、特許請求範囲を含む上記訂正後の明細書を検討し、内容を理解していることをここに表明します。

I hereby state that I have reviewed and understand the contents of the above identified specification, including the claims, as amended by any amendment referred to above.

私は、連邦規則法典第37編第1条56項に定義されるとおり、特許資格の有無について重要な情報を開示する義務があることを認めます。

I acknowledge the duty to disclose information which is material to patentability as defined in Title 37, Code of Federal Regulations, Section 1.56.

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私は、米国法典第35編119条(a)-(d)項又は365条(b)項に基づき下記の、米国外の国の少なくとも一カ国を指定している特許協力条約365(a)項に基づく国際出願、又は外国での特許出願もしくは発明者証の出願についての外国優先権をここに主張するとともに、優先権を主張している、本出願の前に出願された特許または発明者証の外国出願を以下に、枠内をマークすることで、示しています。

Prior Foreign Application(s)

外国での先行出願

<u>11-161329</u>	<u>Japan</u>
(Number)	(Country)
(番号)	(国名)
<u>2000-107072</u>	<u>Japan</u>
(Number)	(Country)
(番号)	(国名)

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(Application No.)
(出願番号)

(Filing Date)
(出願日)

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09/425,788

(Application No.)
(出願番号)

10-22-99

(Filing Date)
(出願日)

(Application No.)
(出願番号)

(Filing Date)
(出願日)

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Priority Not Claimed
優先権主張なし

June 8, 1999

(Day/Month/Year Filed)
(出願年月日)

April 7, 2000

(Day/Month/Year Filed)
(出願年月日)

I hereby claim the benefit under Title 35, United States Code, Section 119(e) of any United States provisional application(s) listed below.

(Application No.)
(出願番号)

(Filing Date)
(出願日)

I hereby claim the benefit under Title 35, United States Code, Section 120 of any United States application(s), or 365(c) of any PCT international application designating the United States, listed below and, insofar as the subject matter of each of the claims of this application is not disclosed in the prior United States or PCT international application in the manner provided by the first paragraph of Title 35, United States Code Section 112, I acknowledge the duty to disclose information which is material to patentability as defined in Title 37, Code of Federal Regulations, Section 1.56 which became available between the filing date of the prior application and the national or PCT international filing date of application.

Pending

(Status: Patented, Pending, Abandoned)
(現況: 特許許可済、係属中、放棄済)

(Status: Patented, Pending, Abandoned)
(現況: 特許許可済、係属中、放棄済)

I hereby declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, under Section 1001 of Title 18 of the United States Code and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

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手続を米特許商標局に対して遂行する弁理士または代理人
として、下記の者を指名いたします。(弁護士、または代理
人の氏名及び登録番号を明記のこと)

POWER OF ATTORNEY: As a named inventor, I hereby appoint
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ること)

(Supply similar information and signature for third and subsequent
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第二共同発明者	日付	Inventor's signature	Date
住所		Residence	
国籍		Citizenship	
私書箱		Post Office Address	
唯一または第一発明者名		Full name of fifth joint inventor, if any	
発明者の署名	日付	Inventor's signature	Date
住所		Residence	
国籍		Citizenship	
私書箱		Post Office Address	
第二共同発明者		Full name of sixth joint inventor, if any	
第二共同発明者	日付	Inventor's signature	Date
住所		Residence	
国籍		Citizenship	
私書箱		Post Office Address	

(第三以降の共同発明者についても同様に記載し、署名をすること)

(Supply similar information and signature for third and subsequent joint inventors.)